

AS2070: Aerospace Structural Mechanics Module 2: Composite Material Mechanics

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Chapters 1-3 in Gibson (2012).

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PRINCIPLES OF

COMPOSITE Material

MECHANICS

Chapters 1-3, 11 in Kollár and Springer (2003).

MECHANICS OF



Chapter 25 in Megson (2013)

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(Also see Daniel and Ishai 2006)

Introduction

- Structural material consisting of multiple non-soluble macro-constituents.
- Main motivation: material properties tailored to applications.
- Both stiffness and strength comes from the fibers/particles, and the matrix holdes everything together.



Types of composite materials (Figure from NPTEL Online-IIT KANPUR (2025))

Examples

- Reinforced concrete
- Wood (lignin matrix reinforced by cellulose fibers)
- Carbon-Fiber Reinforced Plastics (CFRP)

Introduction

- Structural material consisting of multiple non-soluble macro-constituents.
- Main motivation: material properties tailored to applications.
- Both stiffness and strength comes from the fibers/particles, and the matrix holdes everything together.



Continuous fibres

Short fibres/Whiskers

Particulate

Types of composite materials (Figure from NPTEL Online-IIT KANPUR (2025))

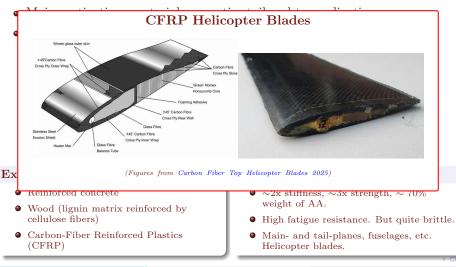
Examples

- Reinforced concrete
- Wood (lignin matrix reinforced by cellulose fibers)
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Introduction

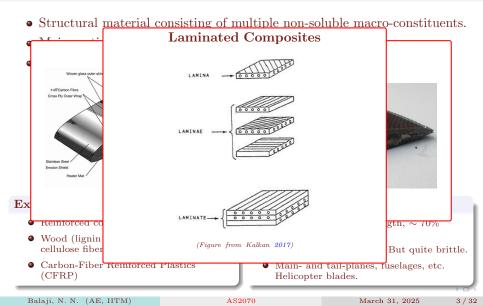
• Structural material consisting of multiple non-soluble macro-constituents.



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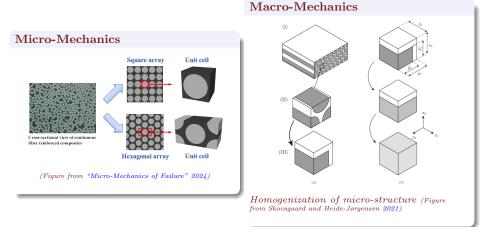
Introduction



1.2. Modeling Composite Material

Introduction

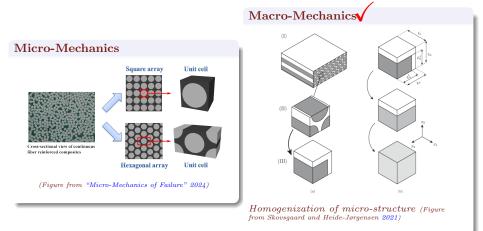
Two main approaches:



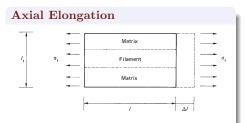
1.2. Modeling Composite Material

Introduction

Two main approaches:



Introduction



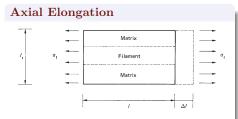
• Strain is fixed, but stress experienced by media differ.

 $\sigma_l = E_l \varepsilon_l$

• Stress-strain relationship simplifies as,

$$\sigma_m = E_m \varepsilon_l, \quad \sigma_f = E_f \varepsilon_l$$
$$\sigma_l A = \sigma_m A_m + \sigma_f A_f$$
$$\implies \boxed{E_l = \frac{A_f}{A} E_f + \frac{A_m}{A} E_m}.$$

Introduction



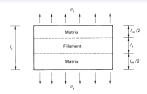
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Transverse Elongation

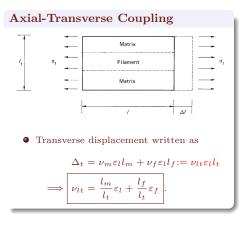


• Stress is fixed, strains differ:

$$\begin{split} \varepsilon_t l_t &= \varepsilon_m l_m + \varepsilon_f l_f \\ \Longrightarrow \frac{\sigma_t}{E_t} l_t &= \frac{\sigma_t}{E_m} l_m + \frac{\sigma_t}{E_f} l_f \\ \Longrightarrow \boxed{\frac{1}{E_t} = \frac{1}{E_m} \frac{l_m}{l_t} + \frac{1}{E_f} \frac{l_f}{l_t}}. \end{split}$$

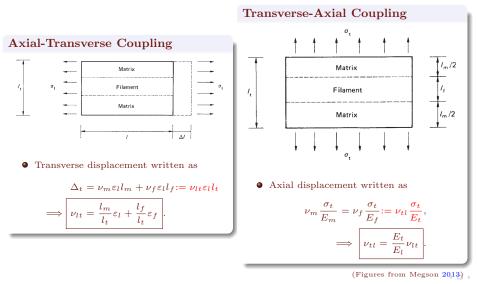
(Figures from	Megson	2013) 🕨
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Introduction: Poisson Effects

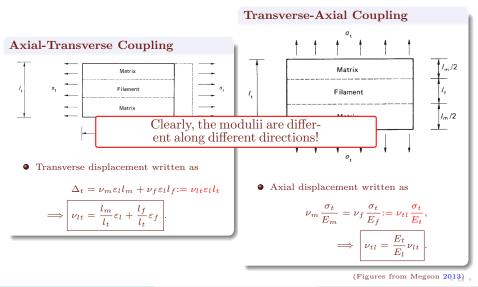


(Figures from Megson 2013)

Introduction: Poisson Effects



Introduction: Poisson Effects



Introduction: Anisotropy

General Anisotropy

σ_{xx}		C_{11}	C_{12}	C_{13}	C_{14}	C_{15}	C_{16}	ε_{xx}
σ_{yy}		C_{12}	C_{22}	C_{23}	C_{24}	C_{25}	C_{26}	ε_{yy}
σ_{zz}	_	C_{13}	C_{23}	C_{33}	C_{34}	C_{35}	C_{36}	ε_{zz}
σ_{xy}	_	C_{14}	C_{24}	C_{34}	C_{44}	C_{45}	C_{46}	γ_{xy}
σ_{xz}		C_{15}	C_{25}	C_{35}	C_{45}	C_{55}	C_{56}	γ_{xz}
σ_{yz}		C_{16}	C_{26}	C_{36}	C_{46}	C_{56}	$\begin{array}{c} C_{16} \\ C_{26} \\ C_{36} \\ C_{46} \\ C_{56} \\ C_{66} \end{array}$	γ_{yz}

Introduction: Anisotropy

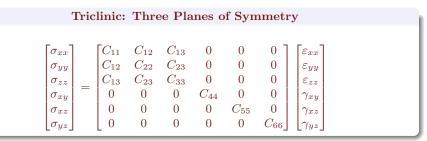
General Anisotropy

$$\begin{bmatrix} \sigma_{xx} \\ \sigma_{yy} \\ \sigma_{zz} \\ \sigma_{xy} \\ \sigma_{xz} \\ \sigma_{yz} \end{bmatrix} = \begin{bmatrix} C_{11} & C_{12} & C_{13} & C_{14} & C_{15} & C_{16} \\ C_{12} & C_{22} & C_{23} & C_{24} & C_{25} & C_{26} \\ C_{13} & C_{23} & C_{33} & C_{34} & C_{35} & C_{36} \\ C_{14} & C_{24} & C_{34} & C_{44} & C_{45} & C_{46} \\ C_{15} & C_{25} & C_{35} & C_{45} & C_{55} & C_{56} \\ C_{16} & C_{26} & C_{36} & C_{46} & C_{56} & C_{66} \end{bmatrix} \begin{bmatrix} \varepsilon_{xx} \\ \varepsilon_{yy} \\ \varepsilon_{zz} \\ \gamma_{xy} \\ \gamma_{xz} \\ \gamma_{yz} \end{bmatrix}$$

Monoclinic: Single Plane of Symmetry

$$\begin{bmatrix} \sigma_{xx} \\ \sigma_{yy} \\ \sigma_{zz} \\ \sigma_{xy} \\ \sigma_{xz} \\ \sigma_{yz} \end{bmatrix} = \begin{bmatrix} C_{11} & C_{12} & C_{13} & C_{14} & 0 & 0 \\ C_{12} & C_{22} & C_{23} & C_{24} & 0 & 0 \\ C_{13} & C_{23} & C_{33} & C_{34} & 0 & 0 \\ C_{14} & C_{24} & C_{34} & C_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & C_{55} & C_{56} \\ 0 & 0 & 0 & 0 & C_{56} & C_{66} \end{bmatrix} \begin{bmatrix} \varepsilon_{xx} \\ \varepsilon_{yy} \\ \varepsilon_{zz} \\ \gamma_{xy} \\ \gamma_{xz} \\ \gamma_{yz} \end{bmatrix}$$

Introduction: Anisotropy

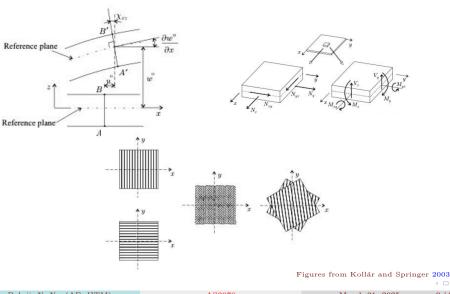


Transversely Isotropic

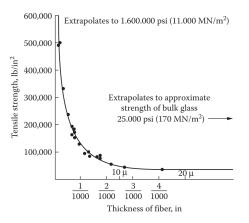
$$\begin{bmatrix} \sigma_{xx} \\ \sigma_{yy} \\ \sigma_{zz} \\ \sigma_{xy} \\ \sigma_{xz} \\ \sigma_{yz} \end{bmatrix} = \begin{bmatrix} C_{11} & C_{12} & C_{13} & 0 & 0 & 0 \\ C_{12} & C_{22} & C_{13} & 0 & 0 & 0 \\ C_{13} & C_{13} & C_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & C_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & C_{44} & 0 \\ 0 & 0 & 0 & 0 & 0 & \frac{C_{11} - C_{12}}{2} \end{bmatrix} \begin{bmatrix} \varepsilon_{xx} \\ \varepsilon_{yy} \\ \varepsilon_{zz} \\ \gamma_{xy} \\ \gamma_{xz} \\ \gamma_{yz} \end{bmatrix}$$

1.4. Classical Laminate Theory

Introduction

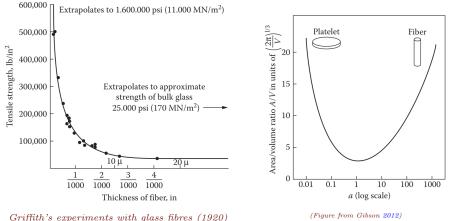


2. Composite Materials



Griffith's experiments with glass fibres (1920) (Figure from Gibson 2012)

2. Composite Materials



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2.1. Types of Composite Materials

Composite Materials

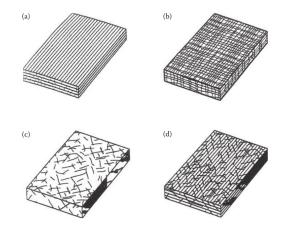


FIGURE 1.4

Types of fiber-reinforced composites. (a) Continuous fiber composite, (b) woven composite, (c) chopped fiber composite, and (d) hybrid composite.

(Figure from Gibson 2012)

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Micro-Mechanics Descriptions

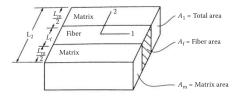
The rule of mixtures is introduced as a very simple framework for developing "overall"/representative mechanical properties.

Basic Definitions

Subscripts $(\cdot)_f$, $(\cdot)_m$, $(\cdot)_v$, and $(\cdot)_c$ denote quantities corresponding to the fiber, matrix, void, and composite (as a whole).

Volume Fraction $v_f = \frac{V_f}{V_c}, v_m = \frac{V_m}{V_c}, v_v = \frac{V_v}{V_c}$ such that $v_f + v_m + v_v = 1$. Note that composite density $\rho_c = \rho_f v_f + \rho_m v_m$.

Weight Fraction $w_f = \frac{\rho_f}{\rho_c} v_f$



(Figure 3.5a from Gibson 2012)

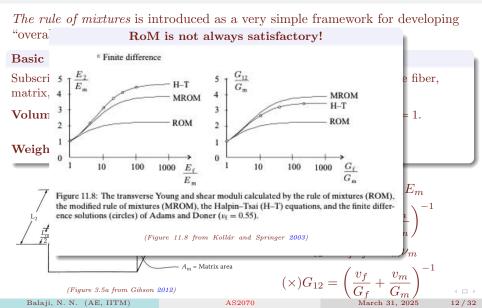
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 $E_{1} = v_{f}E_{f} + v_{m}E_{m}$ $(\times)E_{2} = \left(\frac{v_{f}}{E_{f}} + \frac{v_{m}}{E_{m}}\right)^{-1}$ $\nu_{12} = v_{f}\nu_{f} + v_{m}\nu_{m}$ $(\times)G_{12} = \left(\frac{v_{f}}{G_{f}} + \frac{v_{m}}{G_{m}}\right)^{-1}$ March 31, 2025

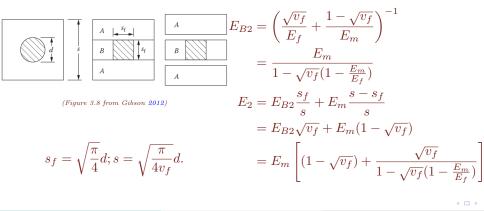
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Micro-Mechanics Descriptions



Micro-Mechanics Descriptions

• The mismatch is related to the fact that our idealized picture was a poor representation of reality to begin with. More geometrical details of the fiber arrangement are necessary.



Micro-Mechanics Descriptions

(Recommended reading: Sec. 3.2.3 in Daniel and Ishai 2006)

The Halpin-Tsai Equation

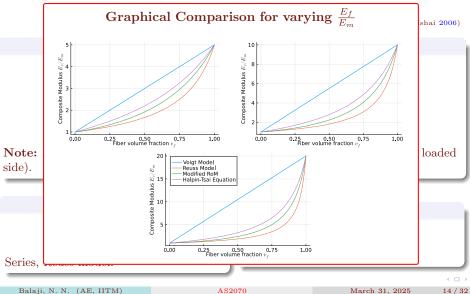
$$E_{2} = E_{m} \frac{1 + \xi \eta v_{f}}{1 - \eta v_{f}}, \quad \eta = \frac{E_{f} - E_{m}}{E_{f} + \xi E_{m}}$$
$$= E_{m} \frac{E_{f} + \xi E_{m} + \xi v_{f} (E_{f} - E_{m})}{E_{f} + \xi E_{m} - v_{f} (E_{f} - E_{m})}$$

Note: $\xi = 2$ for circular section fibers. $\xi = \frac{2a}{b}$ for rectangular fibers (*b* being loaded side).

Case 1:
$$\xi \to 0$$

 $E_2 = \left(\frac{v_f}{E_f} + \frac{1 - v_f}{E_m}\right)^{-1}$
Series, *Reuss* model.
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Micro-Mechanics Descriptions



3.2. Numerical Example

Micro-Mechanics Descriptions

(from Kollár and Springer 2003)

Consider a Graphite/Epoxy unidirectional ply. Matrix properties are given with subscript m in the table below. Nominal properties with fiber volume fraction $v_f = 60\%$ are also given. Assume that the fibers show anisotropy $(E_{f1} \neq E_{f2}).$

	E_1	E_2	G_{12}	ν_{12}	E_m	G_m	ν_m
Value	148	9.65	4.55	0.3	4.1	1.5	0.35

All modulii in GPa.

Estimate the following:

- Fiber modulus properties
- Composite material modulii for volume fraction $v_f = 0.55$.

(Also discussed sensitivity analysis)

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Material Symmetry and Anisotropy

Material Symmetry

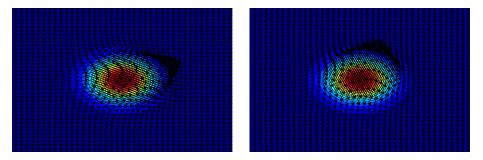
The study of material symmetry is concerned with finding answers to the question: If the strain field on a deformable object is changed, how does the stress field change?

Material Symmetry and Anisotropy

Material Symmetry

The study of material symmetry is concerned with finding answers to the question: If the strain field on a deformable object is changed, how does the stress field change?

Consider the following Deformation Fields



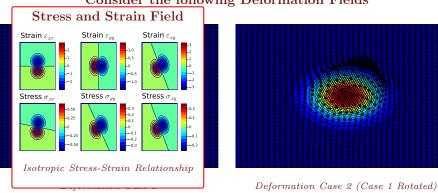
 $Deformation \ Case \ 1$

Deformation Case 2 (Case 1 Rotated)

Material Symmetry and Anisotropy

Material Symmetry

The study of material symmetry is concerned with finding answers to the question: If the strain field on a deformable object is changed, how does the stress field change?

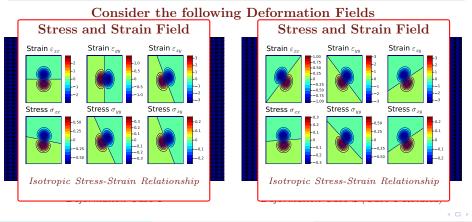


Consider the following Deformation Fields

Material Symmetry and Anisotropy

Material Symmetry

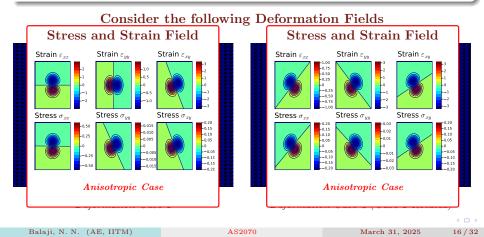
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Material Symmetry and Anisotropy

Material Symmetry

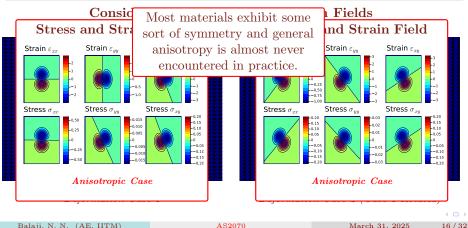
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Material Symmetry and Anisotropy

Material Symmetry

The study of material symmetry is concerned with finding answers to the question: If the strain field on a deformable object is changed, how does the stress field change?



Macro-Mechanics Descriptions

How do stresses and strains transform under coordinate change?

- Suppose $\underline{x} \in \mathbb{R}^3$ are the coordinates of a point in 3D space.
- Let $\underline{x}' \in \mathbb{R}^3$ be the coordinates under transformation.
- We will write: $\underline{x}' = \underline{Q} \underline{x}$, with $\underline{Q}^{-1} = \underline{Q}^T$.

Strains

•
$$\underline{\underline{\varepsilon}} = \frac{1}{2} \left(\underline{\nabla}_{\underline{x}} \underline{u} + \underline{\nabla}_{\underline{x}} \underline{u}^T \right)$$

• $\underline{\nabla}_{\underline{x}'} \underline{u}' = \underline{\underline{Q}} \underline{\nabla}_{\underline{x}} \underline{u} \underline{\underline{Q}}^{-1}$
 $\implies \underline{\underline{\varepsilon}}' = \underline{\underline{Q}} \underline{\underline{\varepsilon}} \underline{\underline{Q}}^T$.

Stresses

• Cauchy Stress Definition: $\underline{t} = \underline{\sigma} \underline{n}$

•
$$\underline{Q} \underline{t} = \underline{t}' = \underline{\underline{\sigma}}' \underline{\underline{n}}' = \underline{\underline{\sigma}}' \underline{Q} \underline{\underline{n}} = \underline{Q} \underline{\underline{\sigma}} \underline{\underline{n}}$$

 $\implies \underline{\underline{\sigma}}' = \underline{Q} \underline{\underline{\sigma}} \underline{Q}^T$

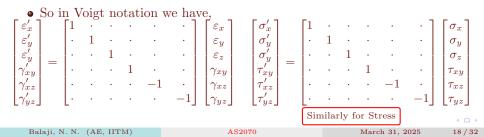
Reflections

Note that reflections may be expressed as a coordinate change with $\underline{\underline{Q}} = \begin{bmatrix} 1 & 0 & 0 \\ 0 & 1 & 0 \\ 0 & 0 & -1 \end{bmatrix}$ (reflection about the *xy* plane).

Macro-Mechanics Descriptions

 $\bullet\,$ Under reflection about the xy plane, the strain transforms as,

$$\begin{bmatrix} \varepsilon'_x & \frac{\gamma'_{xy}}{2} & \frac{\gamma'_{xz}}{2} \\ \varepsilon'_y & \frac{\gamma'_{yz}}{2} \\ \text{sym} & \varepsilon'_z \end{bmatrix} = \begin{bmatrix} 1 & \cdot & \cdot \\ \cdot & 1 & \cdot \\ \cdot & \cdot & -1 \end{bmatrix} \begin{bmatrix} \varepsilon_x & \frac{\gamma_{xy}}{2} & \frac{\gamma_{xz}}{2} \\ \varepsilon_y & \frac{\gamma_{yz}}{2} \\ \text{sym} & \varepsilon_z \end{bmatrix} \begin{bmatrix} 1 & \cdot & \cdot \\ \cdot & 1 & \cdot \\ \cdot & \cdot & -1 \end{bmatrix}$$
$$= \begin{bmatrix} \varepsilon_x & \frac{\gamma_{xy}}{2} & -\frac{\gamma_{xz}}{2} \\ \varepsilon_y & -\frac{\gamma_{yz}}{2} \\ \text{sym} & \varepsilon_z \end{bmatrix}$$



Macro-Mechanics Descriptions

 $\bullet\,$ Under reflection about the xy plane, the strain transforms as,

If a material were symmetric about the xy plane, then reflecting the strain field about the xy plane will result in a stress field that is reflected about the same xy plane.

Note

- Strain field reflection is a <u>kinematic operation</u>/configuration change.
- Change in the Stress field is the <u>effect that the above</u> kinematic change results in.
- If the material happens to be symmetric about the reflection plane, then this change will be a reflection.

 $[\tau'_{yz}]$



 ε'_x ε'_y ε'_y

 $\gamma'_{xy} \\ \gamma'_{xz}$

Similarly for Stress

 σ_x

 σ_y

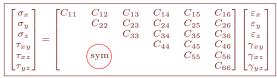
 σ_z

T_{xy} T_{xz}

 τ_{yz}

Macro-Mechanics Descriptions

• We have said the following :



Recall that this symmetry follows from strain energy existence

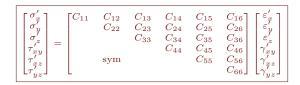
$$\begin{bmatrix} \sigma_{j}^{\prime} \\ \sigma_{y}^{\prime} \\ \sigma_{z}^{\prime} \\ \tau_{xy}^{\prime} \\ \tau_{yz}^{\prime} \\ \tau_{yz}^{\prime} \end{bmatrix} = \begin{bmatrix} C_{11} & C_{12} & C_{13} & C_{14} & C_{15} & C_{16} \\ & C_{22} & C_{23} & C_{24} & C_{25} & C_{26} \\ & & C_{33} & C_{34} & C_{35} & C_{36} \\ & & & C_{44} & C_{45} & C_{46} \\ & & & & & C_{55} & C_{56} \\ & & & & & & C_{66} \end{bmatrix} \begin{bmatrix} \varepsilon_{j}^{\prime} \\ \varepsilon_{j}^{\prime}$$

(The $\underline{\underline{C}}$ matrix is the same in both the original and the reflected coordinate systems)

Macro-Mechanics Descriptions

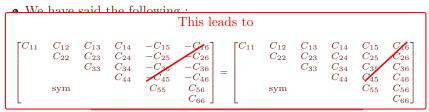
- We have said the following	no •				
This leads to					
$\begin{bmatrix} C_{11} & C_{12} & C_{13} & C_{14} & -C_{15} \\ & C_{22} & C_{23} & C_{24} & -C_{25} \\ & & C_{33} & C_{34} & -C_{35} \\ & & & C_{44} & -C_{45} \\ & & & & & C_{55} \end{bmatrix}$	$-C_{36}$	$ \begin{array}{cccccc} C_{13} & C_{14} & C_{15} & C_{16} \\ C_{23} & C_{24} & C_{25} & C_{26} \\ C_{33} & C_{34} & C_{35} & C_{36} \\ & C_{44} & C_{45} & C_{46} \\ & & C_{55} & C_{56} \\ & & & & C_{66} \end{array} $			

Recall that this symmetry follows from strain energy existence

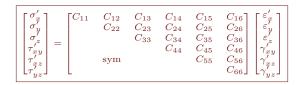


 $(The \underline{\underline{C}} matrix is the same in both the original and the reflected coordinate systems)$

Macro-Mechanics Descriptions

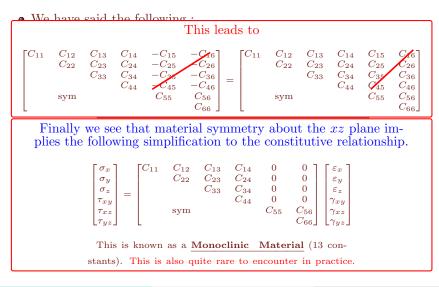


Recall that this symmetry follows from strain energy existence



 $(The \underline{\underline{C}} matrix is the same in both the original and the reflected coordinate systems)$

Macro-Mechanics Descriptions

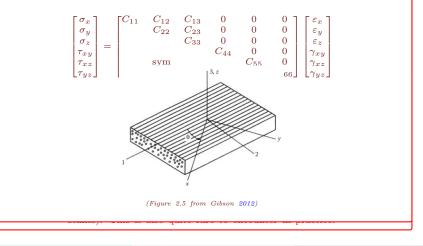


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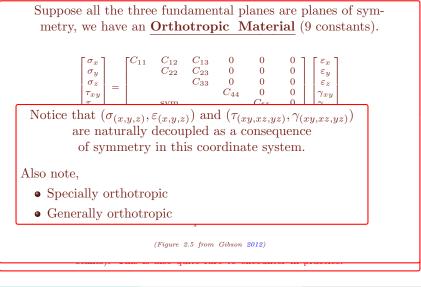
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Macro-Mechanics Descriptions





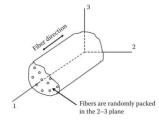
Macro-Mechanics Descriptions



4.1. Material Symmetry and Anisotropy: Transverse Isotropy

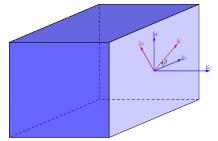
Macro-Mechanics Descriptions

• In continuous fiber reinforced composites, it is often the case that the fibers are randomly distributed on a plane. This leads to planar isotropy in the plane perpendicular to the fiber stacking direction.



⁽Figure 2.6 from Gibson 2012)

• How do the stresses and strains transform on the plane?



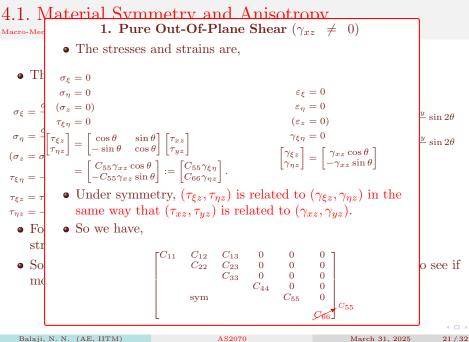
$$\begin{split} & (\sigma_x, \sigma_y, \sigma_z, \tau_{xy}, \tau_{xz}, \tau_{yz}) \to (\sigma_\xi, \sigma_\eta, \sigma_z, \tau_{\xi\eta}, \tau_{\xiz}, \tau_{\etaz}) \\ & (\varepsilon_x, \varepsilon_y, \varepsilon_z, \gamma_{xy}, \gamma_{xz}, \gamma_{yz}) \to (\varepsilon_\xi, \varepsilon_\eta, \varepsilon_z, \gamma_{\xi\eta}, \gamma_{\xiz}, \gamma_{\etaz}) \end{split}$$

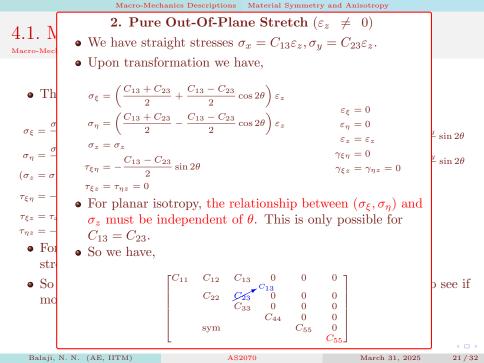
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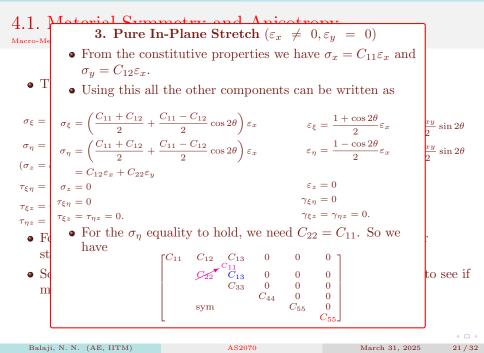
Macro-Mechanics Descriptions

• The stresses and strains transform as follows on the plane:

- For an orthotropic material, the straight stresses/strains and shear stresses/strains are fully decoupled.
- So we will consider different cases of kinematic deformation fields to see if more can be said.





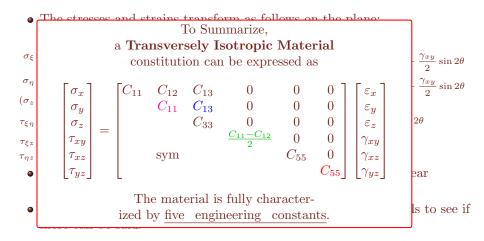


Macro-Mechanics Descriptions

4. Pure In-Plane Shear
$$(\gamma_{xy} \neq 0)$$

• From the constitutive properties we have $\tau_{xy} = C_{44}\gamma_{xy}$.
• Using this all the other components can be written as
 $\sigma_{\xi} = c_{44}\gamma_{xy}\sin 2\theta = C_{11}\varepsilon_{\xi} + C_{12}\varepsilon_{\eta}$ $\varepsilon_{\xi} = \frac{\gamma_{xy}}{2}\sin 2\theta$
 $\sigma_{\eta} = -C_{44}\gamma_{xy}\sin 2\theta = C_{12}\varepsilon_{\xi} + C_{11}\varepsilon_{\eta}$ $\varepsilon_{\eta} = -\frac{\gamma_{xy}}{2}\sin 2\theta$
 $\sigma_{z} = 0$ $\varepsilon_{z} = 0$
 $\tau_{\xi\eta} = C_{44}\gamma_{xy}\cos 2\theta$ $\varepsilon_{z} = 0$
 $\tau_{\xi\tau} = c_{44}\gamma_{xy}\cos 2\theta$ $\gamma_{\xi\eta} = \gamma_{xy}\cos 2\theta$
 $\tau_{\xiz} = \tau_{\eta z} = 0$.
• So we have $C_{44}\gamma_{xy}\sin 2\theta = \frac{C_{11}-C_{12}}{2}\gamma_{xy}\sin 2\theta$. Therefore,
 $\begin{bmatrix} C_{11} & C_{12} & C_{13} & 0 & 0 & 0\\ & C_{13} & 0 & 0 & 0\\ & & C_{33} & 0 & 0 & 0\\ & & & C_{55} \end{bmatrix}$ to see if
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Macro-Mechanics Descriptions



4.1. Material Symmetry and Anisotropy: Engineering Constants

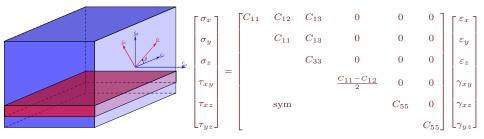
Macro-Mechanics Descriptions

- In engineering practice, the constants are usually written easier in terms of compliance.
- For a specially orthotropic material the strain-stress relationship are usually expressed as,

$$\begin{bmatrix} \varepsilon_x \\ \varepsilon_y \\ \varepsilon_z \\ \gamma_{xy} \\ \gamma_{xz} \\ \gamma_{yz} \end{bmatrix} = \begin{bmatrix} \frac{1}{E_{11}} & -\frac{\nu_{21}}{E_{22}} & -\frac{\nu_{31}}{E_{33}} & 0 & 0 & 0 \\ -\frac{\nu_{12}}{E_{11}} & \frac{1}{E_{22}} & -\frac{\nu_{32}}{E_{33}} & 0 & 0 & 0 \\ -\frac{\nu_{13}}{E_{11}} & -\frac{\nu_{23}}{E_{22}} & \frac{1}{E_{33}} & 0 & 0 & 0 \\ & & & \frac{1}{G_{12}} & 0 & 0 \\ & & & & \frac{1}{G_{13}} & 0 \\ & & & & & \frac{1}{G_{23}} \end{bmatrix} \begin{bmatrix} \sigma_x \\ \sigma_y \\ \sigma_z \\ \tau_{xy} \\ \tau_{xz} \\ \tau_{yz} \end{bmatrix}$$

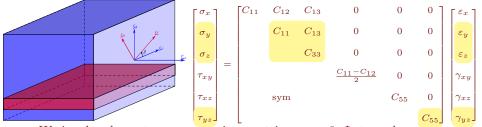
5. Analysis of Planar Laminates

• Let us just consider one thin layer of a transversely isotropic material (continuously reinforced composite along a single direction).



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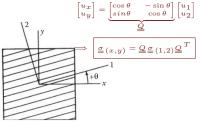


We invoke plane stress assumptions, setting σ_z = 0. Let us also assume small shears, τ_{xz} = 0, τ_{yz} = 0.
(Note: ε_z is not zero, and is implicitly defined)

$$\begin{bmatrix} \sigma_1 \\ \sigma_2 \\ \tau_{12} \end{bmatrix} = \begin{bmatrix} C_{11} & C_{12} & 0 \\ C_{12} & C_{22} & 0 \\ 0 & 0 & C_{33} \end{bmatrix} \begin{bmatrix} \varepsilon_1 \\ \varepsilon_2 \\ \gamma_{12} \end{bmatrix} (4 \text{ constants})$$
(Note change in notation in C_{ij})

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Analysis of Planar Laminates



Positive $\boldsymbol{\theta}$

(Figure 2.11 from Gibson 2012)

$$\begin{split} & \begin{bmatrix} \sigma_x \\ \sigma_y \\ \tau_x y \end{bmatrix} = \underbrace{\begin{bmatrix} \cos^2 \theta & \sin^2 \theta & -2\cos\theta\sin\theta \\ \sin^2 \theta & \cos^2 \theta & 2\cos\theta\sin\theta \\ \cos\theta\sin\theta & -\cos\theta\sin\theta & \cos^2 \theta - \sin^2 \theta \end{bmatrix}}_{\underline{\underline{T}} = 1} \begin{bmatrix} \sigma_1 \\ \sigma_2 \\ \tau_{12} \end{bmatrix} \\ & \underline{\underline{T}} = \begin{bmatrix} \cos^2 \theta & \sin^2 \theta & 2\cos\theta\sin\theta \\ \sin^2 \theta & \cos^2 \theta & -2\cos\theta\sin\theta \\ -\cos\theta\sin\theta & \cos^2 \theta - \sin^2 \theta \end{bmatrix} \end{split}$$

- What if the coordinate system is not aligned with the fiber axes? The stress and strains transform
- In the constitutive relationship we have,

$$\underline{\underline{\sigma}}_{(1,2)} = \underline{\underline{C}} \underline{\underline{\varepsilon}}_{(1,2)}$$

$$\underline{\underline{T}} \underline{\underline{\sigma}}_{(1,2)} = \underline{\underline{\sigma}}_{(x,y)} = \underbrace{\underline{\underline{T}} \underline{\underline{C}} \underline{\underline{T}}^{-1}}_{\underline{\underline{C}}'} \underline{\underline{\varepsilon}}_{(x,y)}$$

where

$$\underline{\underline{C}} = \begin{bmatrix} C_{11} & C_{12} & 0\\ C_{12} & C_{22} & 0\\ 0 & 0 & C_{33} \end{bmatrix}$$

Analysis of Planar Laminates

Analysis of Planar Laminates

$$\begin{bmatrix} u_{y} \\ u_{y} \end{bmatrix} = \begin{bmatrix} \cos \theta & -\sin \theta \\ \sin \theta & \cos \theta \end{bmatrix} \begin{bmatrix} u_{1} \\ u_{2} \end{bmatrix}$$

$$\underbrace{\underline{Q}} \qquad \underbrace{\underline{Q}} \qquad \underbrace{\underline{Q}$$

Analysis of Planar Laminates

• Compliance is often more convenient. This also transforms like \underline{C} so that we have,

• Based on this we can write,

$$\begin{bmatrix} \varepsilon_{y} \\ \varepsilon_{y} \\ \gamma_{xy} \end{bmatrix} = \begin{bmatrix} S_{11}' & S_{12}' & S_{13}' \\ S_{22}' & S_{33}' \\ S_{33}' & S_{33}' \end{bmatrix} \begin{bmatrix} \sigma_{x} \\ \sigma_{y} \\ \tau_{xy} \end{bmatrix}^{-1} \\ S_{11}' = S_{11}c^{4} + S_{22}s^{4} + (S_{33} + 2S_{12})c^{2}s^{2} \\ S_{22}' = S_{11}s^{4} + S_{22}c^{4} + (S_{33} + 2S_{12})c^{2}s^{2} \\ S_{22}' = S_{11}s^{4} + S_{22}c^{4} + (S_{33} + 2S_{12})c^{2}s^{2} \\ S_{33}' = (2S_{11} + 2S_{22} - S_{33} - 4S_{12})2c^{2}s^{2} + S_{33}(c^{4} + s^{4}) \\ S_{12}' = (S_{11} + S_{22} - S_{33})c^{2}s^{2} + S_{12}(c^{4} + s^{4}) \\ S_{13}' = (-2S_{11} + S_{33} + 2S_{12})2c^{3}s + (2S_{22} - S_{33} - 2S_{12})2cs^{3} \\ S_{23}' = (2S_{22} - S_{33} - 2S_{12})2c^{3}s - (2S_{11} - S_{33} - 2S_{12})cs^{3}. \end{bmatrix} \qquad E_{x} = \left[\frac{c^{4}}{E_{1}} + \frac{s^{4}}{E_{2}} + \left(\frac{1}{G_{12}} - \frac{2\nu_{21}}{E_{2}} \right)c^{2}s^{2} \right]^{-1} \\ S_{13}' = (-2S_{11} + S_{33} + 2S_{12})2c^{3}s + (2S_{22} - S_{33} - 2S_{12})2cs^{3}} \\ S_{23}' = (2S_{22} - S_{33} - 2S_{12})c^{3}s - (2S_{11} - S_{33} - 2S_{12})cs^{3}. \end{bmatrix}$$

• In the material principal directions we have, ν_{21} г 1 0 1

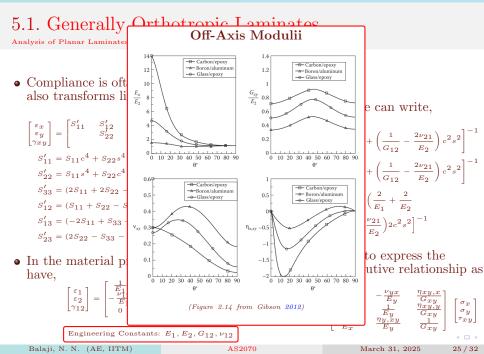
$$\begin{bmatrix} \varepsilon_1\\ \varepsilon_2\\ \gamma_{12} \end{bmatrix} = \begin{bmatrix} \overline{E_1} & -\overline{E_2} & 0\\ -\overline{E_1} & \overline{E_2} & 0\\ 0 & 0 & \overline{d_{12}} \end{bmatrix} \begin{bmatrix} \sigma_1\\ \sigma_2\\ \tau_{12} \end{bmatrix}$$

Engineering Constants: $E_1, E_2, G_{12}, \nu_{12}$
N. N. (AE, IITM)

• It is customary to express the laminate constitutive relationship as

$$\begin{bmatrix} \varepsilon_x \\ \varepsilon_y \\ \gamma_{xy} \end{bmatrix} = \begin{bmatrix} \frac{1}{E_x} & -\frac{\nu_{yx}}{E_y} & \frac{\eta_{xy,x}}{G_{xy}} \\ -\frac{\nu_{xy}}{E_x} & \frac{1}{E_y} & \frac{\eta_{xy,y}}{G_{xy}} \\ \frac{\eta_{x,xy}}{E_x} & \frac{\eta_{y,xy}}{E_y} & \frac{1}{G_{xy}} \end{bmatrix} \begin{bmatrix} \sigma_x \\ \sigma_y \\ \tau_{xy} \end{bmatrix}$$

$$(\Box \models)$$
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5.2. Numerical Examples: 1

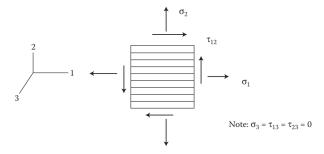
Analysis of Planar Laminates(Example 2.2 from Gibson 2012)

Consider an orthotropic laminate with the properties

 $E_1 = 140 \text{ GPa}, E_2 = 10 \text{ GPa}, G_{12} = 7 \text{ GPa}, \nu_{12} = 0.3, \nu_{23} = 0.2.$

Compute the strains if it is subjected to the following state of stress in the principal coordinates:

 $\sigma_1 = 70 \text{ MPa}, \sigma_2 = 140 \text{ MPa}, \tau_{12} = 35 \text{ MPa}, \sigma_3 = \tau_{12} = \tau_{23} = 0.$



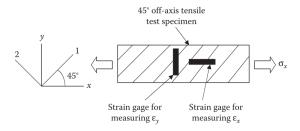
(Figure 2.10 from Gibson 2012)

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5.2. Numerical Examples: 2

Analysis of Planar Laminates(Example 2.3 from Gibson 2012)

A 45° off-axis tensile test is conducted on a generally orthotropic test specimen by applying a normal stress σ_x . The specimen has strain gauges attached to measure axial and transverse strains ($\varepsilon_x, \varepsilon_y$). How many engineering parameters can be estimated from measurements of $\sigma_x, \varepsilon_x, \varepsilon_y$?



(Figure 2.15 from Gibson 2012)

6. Classical Laminate Theory

• In the Kirchhoff-Love Plate Theory we had,

$$\begin{bmatrix} \underline{\underline{N}} \\ \underline{\underline{M}} \end{bmatrix} = \begin{bmatrix} \underline{\underline{A}} & \underline{\underline{B}} \\ \underline{\underline{\underline{B}}} & \underline{\underline{\underline{D}}} \end{bmatrix} \begin{bmatrix} \underline{\underline{u}'} \\ \underline{\underline{w}''} \end{bmatrix}$$

where

$$\underline{\underline{A}} = \frac{Et}{1-\nu^2} \begin{bmatrix} 1 & \nu & 0\\ \nu & 1 & 0\\ 0 & 0 & \frac{1-\nu}{2} \end{bmatrix}, \quad \underline{\underline{D}} = \frac{Et^3}{12(1-\nu^2)} \begin{bmatrix} 1 & \nu & 0\\ \nu & 1 & 0\\ 0 & 0 & \frac{1-\nu}{2} \end{bmatrix}, \quad \underline{\underline{B}} = \underline{\underline{0}}.$$

• This can also be written in terms of thickness moments of the constitutive matrix $\underline{\underline{C}} = \frac{E}{1-\nu^2} \begin{bmatrix} 1 & \nu & 0 \\ \nu & 1 & 0 \\ 0 & 0 & \frac{1-\nu}{2} \end{bmatrix}$ as $\underline{\underline{A}} = \int_{-\frac{t}{2}}^{\frac{t}{2}} \underline{\underline{C}} dz, \quad \underline{\underline{B}} = \int_{-\frac{t}{2}}^{\frac{t}{2}} z \underline{\underline{C}} dz, \quad \underline{\underline{D}} = \int_{-\frac{t}{2}}^{\frac{t}{2}} z^2 \underline{\underline{C}} dz.$

6. Classical Laminate Theory

- Suppose we had different laminate plies along the thickness, such that the constitutive matrix is $\underline{\underline{C}}_i$ for $z \in (z_i, z_{i+1})$ and $-\frac{t}{2} = z_1 < \cdots < z_N = \frac{t}{2}$.
- Then the A B D matrices are written as the sums,

$$\underline{\underline{A}} = \sum_{i} (z_{i+1} - z_i) \underline{\underline{C}}_i, \quad \underline{\underline{B}} = \sum_{i} \frac{z_{i+1}^2 - z_i^2}{2} \underline{\underline{C}}_i, \quad \underline{\underline{D}} = \sum_{i} \frac{z_{i+1}^3 - z_i^3}{3} \underline{\underline{C}}_i.$$

- Unlike isotropic plates, composite laminates can have non-zero $\underline{\underline{B}}$ matrix (moment-planar coupling), bending-twisting coupling, etc.
- This $\begin{bmatrix} \underline{\underline{A}} & \underline{\underline{B}} \\ \underline{\underline{\underline{B}}} & \underline{\underline{\underline{D}}} \end{bmatrix}$ matrix is known as the **Laminate Compliance Matrix**.

6.1. The Laminate Orientation Code

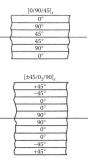
Classical Laminate Theory

- Ply angles separated by slashes, ordered from top to bottom
- Subscript "s" for symmetric laminates
- Numerical subscripts for repetitions
- Center ply with an overbar for odd laminates

(See sec. 7.1 in Gibson 2012)

Types

- Symmetric, Antisymmetric, Asymmetric
- Angle-Ply, Cross-Ply, Balanced, $\pi/4$ laminates



	$[(0/\mp 30)_2]$	
	0°	7
	-30°	
	+30°	7
	0°	
L	-30°	5
	+30°	





[90	/45/30/0/90	/45]
	90°	7
	45°	
)	30°	7
	0°	
	90°	
	45°	



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6.1. The Laminate Orientation Code Summary of Laminate Stiffnesses

Classical Laminate Theory

Table 3.4. The [A], [B], [D] matrices for laminates. When the laminate is summatrical the [B] matrix is zero. Cross plu laminates are orthotronic

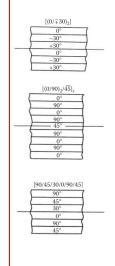
- Ply angles separ ordered from to
- Subscript "s" fo laminates
- Numerical subs repetitions
- Center ply with odd laminates

(See Typ

• Symmetric, Antis Asymmetric

• Angle-Ply, Crosslaminates

[A]		[<i>B</i>]	[D]	
Symmetri	cal			
	2 A16	[0 0 0]	$\begin{bmatrix} D_{11} & D_{12} \\ D_{12} & D_{22} \\ D_{16} & D_{26} \end{bmatrix}$	D_{16}
A12 A	2 A26	$\begin{bmatrix} 0 & 0 & 0 \\ 0 & 0 & 0 \\ 0 & 0 & 0 \end{bmatrix}$	D_{12} D_{22}	D26
A_{16} A_{16}	6 A 66	0 0 0	D_{16} D_{26}	D_{66}
Balanced				
	2 0]	$\begin{bmatrix} B_{11} & B_{12} & B_{13} \end{bmatrix}$	$\begin{bmatrix} D_{11} & D_{12} \end{bmatrix}$	D_{16}
A12 A	12 0	B12 B22 B2	$\begin{bmatrix} D_{11} & D_{12} \\ D_{12} & D_{22} \\ D_{16} & D_{26} \end{bmatrix}$	D26
0 0	A 66	B_{16} B_{26} B_{6}	$[D_{16} D_{26}]$	D 66
Orthotro	pic			
	2 0]	$\begin{bmatrix} B_{11} & B_{12} & 0 \end{bmatrix}$	$\begin{bmatrix} D_{11} & D_{12} \end{bmatrix}$	0]
A12 A	2 0	B_{12} B_{22} 0	$\begin{bmatrix} D_{11} & D_{12} \\ D_{12} & D_{22} \\ 0 & 0 \end{bmatrix}$	0
0 0	A_{66}	$0 0 B_6$	6 0 0	D ₆₆
Isotropic				
$\begin{bmatrix} A_{11} & A_{1} \end{bmatrix}$	2 0]	$\begin{bmatrix} B_{11} & B_{12} \end{bmatrix}$	$0] \begin{bmatrix} D_{11} & D_{12} \end{bmatrix}$	0
A12 A	1 0	B_{12} B_{11}	$\begin{bmatrix} 0 \\ 0 \\ \frac{1-B_{12}}{2} \end{bmatrix} \begin{bmatrix} D_{11} & D_{12} \\ D_{12} & D_{11} \\ 0 & 0 \end{bmatrix}$	0
0 0	$\frac{A_{11}-A_{12}}{2}$	0 0 81	$\left[\frac{1-B_{12}}{2}\right] = 0 = 0$	$\frac{D_{11}-D_{12}}{2}$
Quasi-iso	tropic			
	2 0]	$\begin{bmatrix} B_{11} & B_{12} & B_{1} \end{bmatrix}$	$\begin{bmatrix} D_{11} & D_{12} \end{bmatrix}$	D_{16}
A12 A	1 0	B12 B22 B	$D_{12} D_{22}$	D26
0 0	$\frac{A_{11} - A_{12}}{2}$	B16 B26 B	$\begin{bmatrix} D_{11} & D_{12} \\ D_{12} & D_{22} \\ D_{16} & D_{26} \end{bmatrix}$	D 66



Gibson 2012)

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6.2. Laminated Beams

Classical Laminate Theory

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